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Many reduced niobium oxides in A-Nb-O ternary systems where A is an alkaline or alkaline earth metal have been reported. Some examples are LiNbO2 (1), NaNb<sub>10</sub>O<sub>18</sub> (2), BaNb<sub>8</sub>O<sub>14</sub> (3), Ba<sub>2</sub>Nb<sub>15</sub>O<sub>32</sub> (4) and Ca<sub>0.75</sub>Nb<sub>3</sub>O<sub>6</sub> (5). Some of strongly reduced niobates contain a [Nb6O12]O6 cluster with metal-metal bonding as found in NaNb10O18 (2), BaNb8O14 (3), Ba2Nb15O32 (4) and These cluster compounds seemed like particularly good Rb4Al2Nb36O70 (6). candidates in view of the known superconducting Mo-S cluster compounds. Furthermore, it was reported that LiNbO2 changed from a semiconductor to a superconductor with a Tc = 4.5 K by deintercalation of Li<sup>+</sup> ions (7). This confirms the potential for superconductivity in reduced niobates. Recently, we discovered a new reduced rubidium niobium oxide, Rb1.51Nb10O17, with a layered structure and having a [Nb6O12]O6 cluster (8). This compound is not superconducting down to 4.2 K, but changing the electron concentration in the cluster seemed like The Rb+ ions occupying the interlayer space appeared to be a worthwhile goal. good candidates for deintercalation or ion exchange reactions.

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# **Exploratory Synthesis Employing Unusual Synthesis Approaches**

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### Introduction

One of the areas we investigated for new superconductors is reduced oxides of niobium. One unusual synthesis approach to prepare these materials has been ion exchange reactions and another has been quenching reactions from very high temperatures.

Many reduced niobium oxides in A-Nb-O ternary systems where A is an alkaline or alkaline earth metal have been reported. Some examples are LiNbO2 (1), NaNb<sub>10</sub>O<sub>18</sub> (2), BaNb<sub>8</sub>O<sub>14</sub> (3), Ba<sub>2</sub>Nb<sub>15</sub>O<sub>32</sub> (4) and Ca<sub>0.75</sub>Nb<sub>3</sub>O<sub>6</sub> (5). Some of strongly reduced niobates contain a [Nb6O12]O6 cluster with metal-metal bonding as found in NaNb10O18 (2), BaNb8O14 (3), Ba2Nb15O32 (4) and These cluster compounds seemed like particularly good Rb4Al2Nb36O70 (6). candidates in view of the known superconducting Mo-S cluster compounds. Furthermore, it was reported that LiNbO2 changed from a semiconductor to a superconductor with a Tc = 4.5 K by deintercalation of Li<sup>+</sup> ions (7). This confirms the potential for superconductivity in reduced niobates. Recently, we discovered a new reduced rubidium niobium oxide, Rb1.51Nb10O17, with a layered structure and having a [Nb6O12]O6 cluster (8). This compound is not superconducting down to 4.2 K, but changing the electron concentration in the cluster seemed like The Rb+ ions occupying the interlayer space appeared to be good candidates for deintercalation or ion exchange reactions.

## **Cluster Compounds**

Powder samples of Rb<sub>1.51</sub>Nb<sub>10</sub>O<sub>17</sub> were obtained by heating a mixture of Rb<sub>8</sub>Nb<sub>22</sub>O<sub>59</sub> (9) and Nb powder with the molar ratio 1:14 in an evacuated quartz tube at 1050°C for 12 hr. Powder of Rb<sub>8</sub>Nb<sub>22</sub>O<sub>59</sub> was prepared by heating a stoichiometric mixture of Rb<sub>2</sub>CO<sub>3</sub> and Nb<sub>2</sub>O<sub>5</sub> at 1000°C for 15 hr. The products always contained a small amount of NbO and NbO<sub>2</sub>. The hexagonal lattice constants for the powder sample are a = 6.021(2) and c = 38.90(1)Å, agreeing well with those reported (8) for a single crystal (a = 6.020(1) and c = 38.703(8)Å).

Ion-exchange reactions were performed with various alkali chloride and HCl solutions. Deintercalation reactions were performed with aqueous KMnO4 solution and with I2 or Br2 solutions in acetonitrile. The reaction temperature ranged from room temperature to 180°C. For reactions above 100°C, an autoclave (70ml) with a teflon lining was used as a reaction vessel. The reaction duration was 1-10 days. The solid products were separated by centrifuging, washed with distilled water or acetonitrile, and dried at 50°C.

The products were identified by X-ray powder diffraction using CuKα radiation. Lattice constants were determined by using silicon as an internal standard. Products were completely decomposed by a concentrated HCl solution under hydrothermal conditions at 180°C for 15 hr to yield a white amorphous powder which crystallized to Nb<sub>2</sub>O<sub>5</sub> at elevated temperatures. After filtration, the amount of alkali in the solution was analyzed by atomic absorption spectroscopy.

Neither ion-exchanged nor deintercalation was detected below 100°C. However under hydrothermal conditions above 100°C, rubidium ions could be partially ion-exchanged and deintercalated. The starting material was completely oxidized to white amorphous powder in concentrated acid at temperatures above about 150°C. The X-ray powder patterns for ion-exchanged and deintercalated compounds are indexed with hexagonal cells similar to that of the starting material, and their lattice constants are slightly changed when compared with those of the starting material. This fact suggests that the crystal structure of products remains intact during this soft-chemical reaction. TG curves and gas evolution curves were measured in a He atmosphere and in air for the Li+ and K+ ion-exchanged compounds. Weight gain caused by oxidation of niobium is observed above about 300°C in the TG curves for the ion-exchanged and deintercalated compounds measured in air; whereas in He atmosphere, no weight gains are observed. A large evolution of H<sub>2</sub>O gas up to about 150°C and small continuous one up to about 500°C are observed for all the ion-exchanged compounds. A distinct H<sub>2</sub>O evolution at about 400°C is observed for the H<sup>+</sup>, Li<sup>+</sup> and Cs+ ion-exchanged compounds. No gas evolution is observed for the deintercalated compound. Table 1 shows chemical composition, lattice constants and weight changes for the ion-exchanged and deintercalated compounds. amount of H2O was estimated from the weight loss of TG curves measured in He atmosphere.

The maximum replacement of Rb+ ions was observed for the compound ionexchanged with K+ ions at 180°C for 5 days. The sum of rubidium and other alkali ions for all the alkali ion-exchanged compounds is less than the amount of rubidium ion (1.51) in the starting material. Two explanations were considered. Either niobium is oxidized with a decrease of cations in the interlayer space or protons are incorporated to maintain the niobium oxidation state during the ionexchange reaction. The latter explanation appears more likely partly because of the distinct evolution of H2O at about 400°C for the H+, Li+ and Cs+ ionexchanged compounds. This water is thought to be formed by protons and oxygen atoms of the host lattice, such protons being incorporated into the niobate during the ion-exchange reactions with alkaline ions. For the Na+ exchanged compounds, no distinct evolution of H2O gas is observed at about 400°C. Instead, there is a gradual weight loss and small continuous evolution of H2O gas up to about 500°C, following to the removal of water in the interlayer space. This suggests that the Na+ and K+ ion-exchanged compounds also incorporate protons during the ion-exchange reaction. The weight gain observed in TG curves measured in air is always smaller than the calculated one, assuming no oxidation. The niobium in the ion-exchanged compounds may be partially oxidized during the ion-exchange reactions. However, the change of oxidation state of niobium during soft-chemical reaction cannot be precisely estimated because of the NbO and NbO2 impurities in the reactant ion exchanged.

TABLE 1
Chemical composition, lattice constants and weight changes for Rb1.51Nb10O17, ion-exchanged and deintercalated compounds

Chemical composition	a (Å)	c(Å)	(1)	(2)	(3)
Rb <sub>1.51</sub> Nb <sub>10</sub> O <sub>17</sub>	6.021(2)	38.90(1)		9.74	10.53
$Rb_{0.25}H_{1.25}Nb_{10}O_{17}\cdot 0.8H_2O$	5.970(8)	40.62(5)	2.09	8.12	8.43
Li <sub>0.67</sub> Rb <sub>0.31</sub> H <sub>0.53</sub> Nb <sub>10</sub> O <sub>17</sub> ·0.8H <sub>2</sub> O	6.005(8)	40.77(5)	1.58	8.94	9.35
Na0.86Rb0.28H0.37Nb10O17·2.5H2O	6.007(9)	40.73(5)	3.69	3.91	6.88
1va0.86ftb0.28f10.37fvb10O17-2.5f12O	0.007(3)	40.73(3)	5.05	0.91	0.00
$K_{1.02}Rb_{0.02}H_{0.47}Nb_{10}O_{17}\cdot 0.9H_{2}O$	5.998(7)	40.98(6)	1.62	7.97	9.20
Cs <sub>0.44</sub> Rb <sub>0.48</sub> H <sub>0.59</sub> Nb <sub>10</sub> O <sub>17</sub> ·0.8H <sub>2</sub> O	5.996(8)	40.78(6)	1.54	5.99	8.79
Rb <sub>0.15</sub> Nb <sub>10</sub> O <sub>17</sub>	6.018(2)	39.16(2)		8.70	10.64

- (1) Weight loss(%) measured in He atmosphere.
- (2) Total weight gain(%) measured in air.
- (3) Calculated total weight gain(%).

Mass spectrometry shows no gas evolution for the deintercalated compounds, indicating the reduced negative charge on the niobate layer is compensated only by the Rb+ ions in the interlayer space. Deintercalation proceeds to Rb0.15Nb10O17 when treated with 0.21M Br2 acetonitrile solution at 120°C for 7 days. As seen in Table 1, the lattice constants of the oxidized products change only slightly during the deintercalation. During deintercalation of layered AMO2 oxides (A = alkali metals and M = transition metals), a more significant increase in the lattice spacing perpendicular to the layers is frequently noted and is attributed to increased repulsion between these layers. Because the niobium ions in the NbO6 octahedra and the NbO4 tetrahedra are presumably in the fully oxidized +5 state, oxidation during deintercalation only occurs in the [Nb6O12]O6 cluster. This cluster is shielded from the region of alkali cations by the NbO<sub>6</sub> octahedra and the NbO<sub>4</sub> tetrahedra. This may partly explain the smallness of the expansion along the c axis. The formal oxidation state of Nb in the octahedral Nb6 cluster is calculated to increase from 2.08 to 2.31 during deintercalation. This oxidation of the cluster is expected to change interatomic distance within the cluster, and cluster orientation may also change. factors will also influence the lattice parameters of the product of deintercalation.

The a.c. susceptibility of materials prepared was examined down to 4.2 K using a Lakeshore AC Susceptometer. No evidence of superconductivity in our products of ion-exchanging or deintercalating Rb1.51Nb10O17 was observed down to 4.2 K. The Rb+ ions in the reduced niobium oxide Rb1.51Nb10O17 were partially ion-exchanged by other alkali ions. Simultaneous incorporation of protons occurred. Deintercalation of Rb1.51Nb10O17 with a Br2 solution in acetonitrile at

120°C yielded Rb<sub>0.15</sub>Nb<sub>10</sub>O<sub>17</sub>. Unlike the case of LiNbO<sub>2</sub>, this deintercalation reaction does not lead to a product showing superconductivity above 4.2 K.

# **High Temperature Synthesis**

A Brew furnace was used to achieve temperatures as high as 2000°C followed by rapid quenching to room temperature. This was a highly exploratory approach seeking entirely new compounds that might be superconducting or have other interesting properties. Systems investigated were Mg/Nb/O, Ca/Nb/O, Sr/Nb/O, Ba/Nb/O, Al/Nb/O, Cr/Nb/O, Ti/Nb/O, La/Nb/O, Ce/Nb/O, Sc/Nb/O, and Y/Nb/O. Syntheses were conducted in high vacuum to insure reducing conditions. No new superconducting phases were discovered.

Miscellaneous Incomplete Results

Superconductivity was recently discovered at low temperatures in SrRuO<sub>3</sub>. This suggests that other oxides containing Ru should be investigated for possible superconductivity. New phases were prepared which contain Ba, Ru, O, and either W, Mo, V, or Cr. We continue to investigate the structure and properties of these new compounds. In related work, we have for the first time grown crystals of BaPtO<sub>3</sub>.

Superconductivity occurs in the  $BaBiO_3$ -  $KBiO_3$  solid solution. The compound  $LiBiO_3$  was not previously known, but we have now prepared it using hydrothermal conditions. The structure has been determined and found to be completely different than that of  $KBiO_3$ .

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